

# **Appendix 1**

## **List of participants and excused persons**

**Participants EFC WP15 meeting 6<sup>th</sup> September 2005 Lisbon**

<b>Surname</b>	<b>Name</b>	<b>Company</b>	<b>Country</b>
Martin	Richez	Total	FRANCE
Stuart	Bond	TWI	UK
Chris J	Claesen	Nalco	BELGIUM
Hennie	de Bruyn	Borealis AS	NORWAY
Paul	Eaton	Champion Technologies Inc	USA
Alec	Groysman	Oil Refineries Ltd	ISRAEL
Craig A.	Howard	GE Infrastructure	AUSTRALIA
Joanna	Hucinska	Gdansk Technical University	POLAND
Russell	Kane	Intercorr	USA
Maarten	Lorenz	Shell Global Solutions International B.V.	NETHERLANDS
Ellina	Lunarska	Institute of Physical Chemistry	POLAND
David	Owen	GE Betz	UK
Andrew M	Pritchard	Corrosion & Fouling Consultancy	UK
John	Pugh	Innovene Grangemouth	UK
Roberto	Riva	Eni Technologie	ITALY
Francois	Ropital	Institut Français du Pétrole	FRANCE
Rob	Scanlan	Conoco	UK
Günter	Schmitt	Lab for Corrosion Protection	GERMANY
Laszlo	Simor	Danube Refinery	HUNGARY
Nick	Smart	Sercos Assurance F	UK
Stefano	Trasatti	University of Milan	ITALY
Stefan	Winnik	Exxon Mobil Chemical	UK

**Excuses received for the EFC WP15 meeting 6<sup>th</sup> September 2005 Lisbon**

<b>Surname</b>	<b>Name</b>	<b>Company</b>	<b>Country</b>
Curt	Christensen	Force Institutes	DENMARK
André	Claus	GE Betz	BEGIUM
Michael	Davies	CARIAD Consultants	GREECE
Nicholas	Dowling	Shell Global Solutions International B.V.	NETHERLANDS
Charles	Droz	Exxon Mobil	FRANCE
Sebastien	Duval	Saipem	FRANCE
Carlo	Farina	Corrosion Consultant	ITALY
Tiina	Hakonen	FORTUM Oil & Gas Oy	FINLAND
Martin	Holmquist	AB Sandvik Steel	SWEDEN
Andrew	Kettle	Chevron Texaco Ltd	UK
Mario	Lanciotti	Polimeri Europa S.p.A	ITALY
Morten	Langøy	Bodycote Materials Testing AS	NORWAY
Istvan	Lukovits	Chemical Research Center	HUNGARY
Richard	Pargeter	TWI	UK
Kirsi	Rintamaki	FORTUM Oil & Gas Oy	FINLAND
Iris	Rommerskirchen	Butting Edelstahlwerke GmbH&Co KG	GERMANY
Liane	Smith	Intetech Ltd	UK
Betrand	Szymkowiak	IFP Technology Group - AXENS	FRANCE
John	Thirkettle	Thor Corrosion	UK
Alan	Turnbull	National Physical Laboratory	UK

# **Appendix 2**

## **EFC WP15 Activities**

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- **Presentation of the activities of WP15**
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## European Federation of Corrosion (EFC)

- Federation of 32 National Associations
- 19 Working Parties (WP) + 1 Task Force
- Annual Corrosion congress « Eurocorr »
- Thematic workshops and symposiums
- Working Party meetings (for WP15 twice a year)
- Publications
- EFC - NACE agreement
- for more information <http://www.efcweb.org>

WP 15 meeting September 6<sup>th</sup> 2005 Lisbon . . . . .

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## EFC Working Parties

- WP 1: Corrosion Inhibition
- WP 3: High Temperature
- WP 4: Nuclear Corrosion
- WP 5: Environmental Sensitive Fracture
- WP 6: Surface Science and Mechanisms of corrosion and protection
- WP 7: Education
- WP 8: Testing
- WP 9: Marine Corrosion
- WP 10: Microbial Corrosion
- WP 11: Corrosion of reinforcement in concrete
- WP 12: Computer based information systems
- WP 13: Corrosion in oil and gas production
- WP 14: Coatings
- WP 15: Corrosion in the refinery industry
- WP 16: Cathodic protection
- WP 17: Automotive
- WP 18: Tribocorrosion
- WP 19: Corrosion of polymer materials
- Task Force 2: Corrosion and Protection of steel structures

WP 15 was created in sept. 96 with J. Harston as first chairman

WP 15 meeting September 6<sup>th</sup> 2005 Lisbon . . . . .

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## EFC Working Party 15 « Corrosion in Refinery » Activities

The following are the main areas being pursued by the Working Party:

· Information Exchange

- Sharing of refinery materials /corrosion experiences by operating company representatives.

· Forum for Technology

- Sharing materials/ corrosion/ protection/ monitoring information by providers

Publications

*WP 15 meeting September 6<sup>th</sup> 2005 Lisbon* • • • • • • • •

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## EFC Working Party 15 « Corrosion in Refinery » Activities

WP Meetings

· One WP 15 working party meeting in Spring,  
(this year on 17-18 March 2005 in Trondheim)

· One meeting at Eurocorr in conjunction with the conference,  
(this meeting during Eurocorr 2005 4-8 September in Lisbon)

Eurocorr Conference sessions (September)

Refinery Corrosion Session

+ Workshops or Joint Session with other EFC WP parties

WP15 page in EFC Web site

[http://www.efcweb.org/WP\\_on\\_Corrosion\\_in\\_the\\_Refinery\\_Industry.html](http://www.efcweb.org/WP_on_Corrosion_in_the_Refinery_Industry.html)

*WP 15 meeting September 6<sup>th</sup> 2005 Lisbon* • • • • • • • •



EUROPAISCHE FÖDERATION KORROSION  
 EUROPEAN FEDERATION OF CORROSION  
 FEDERATION EUROPEENNE DE LA CORROSION



[ENGLISH](#) | [SITE MAP](#) | [CONTACT](#) | [RUSSIAN](#) | [OUR WEBSITES](#)

- Home
- Structure
- Member Societies
- Working Parties
  - WP on Corrosion Education
  - WP on Corrosion by Hot Gases and Combustion Products
  - WP on Corrosion in Oil and Gas Production
  - WP on Corrosion in the Refinery Industry
  - WP on Automotive Corrosion
- Board of Administrators
- ETAC
- General Secretariat
- Events and Highlights
- Offers
- News
- EFC/NACE

## EFC Working Party 15: Corrosion in the Refinery Industry

### Vision:

The Working Party meetings objectives are:

- Information Exchange
  - Sharing of refinery materials/corrosion/inspection experiences by operating company representatives.
- Forum for Technology
  - Sharing materials/corrosion/protection/monitoring information by providers, users, R&D.
- Scientific exchange
  - Sharing materials/corrosion/protection scientific works.
- Development of documents, guidelines, publications related to corrosion in the refinery industry.

### Strategy Plan:

- Survey of corrosion problems in refinery industry

The group will collect information first on hydrotreatment and hydrocracking units, then processes as FCC, Catalytic reforming, Distillation, Sulfur plant, Alkylation, Sourwater stripper will be considered. Guideline and publication will be issued.

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### Excerpt of last meeting minutes

[Minutes of the 15th September 2004 meeting and Appendices](#)

[Minutes of the 8-9th March 2004 meeting, Appendices 1-7, Appendices 8-11, and Appendix Literature](#)

[Minutes of the 30th September 2003 meeting and Appendix](#)

[Minutes of the 10th April 2002 meeting](#)

[Minutes of the 15th November 2002 meeting](#)

### Agenda of forthcoming meetings

WP 15 meeting on March 17-18<sup>th</sup> in Trondheim, Norway  
 The agenda is not completely finalized but one day will be dedicated to general topics and the other one will be dedicated to the guideline on Corrosion Under Insulation.

### Publications

- EFC Publication No. 40 "Requirements for cooling water systems"
- EFC Publication No. 42 "Collection of Selected Papers"

Publications are in preparation on the following subjects:

- Survey on amine unit corrosion
- Guideline on corrosion under insulation

WP 15 meeting September 6<sup>th</sup> 2005 Lisbon

## **Appendix 3**

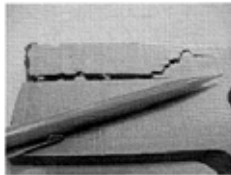

# **Proposal of a typical failure refinery corrosion cases atlas**



Typical refinery failure cases atlas ?

CORROSION ATLAS		CASE HISTORY
		01.06.20.04
MATERIAL	Industrial steel	
SYSTEM	Carbon dioxide scrubbing tower	
PART	WALL	
PHENOMENON	Cold hydrogen attack	
		
APPEARANCE	Formation of carbon in the steel	
TIME TO FAILURE	More years	
ENVIRONMENT	Water with dissolved CO <sub>2</sub>	
CAUSE	Influx in the steel of atomic hydrogen formed by slight corrosion. The atomic hydrogen recombines in the steel, forms CH <sub>4</sub> and eventually CH <sub>2</sub> . Because of the high pressure the CH <sub>4</sub> is expelled and the CH <sub>2</sub> remains in the steel.	
REMEDY	<ul style="list-style-type: none"> <li>Changeover to another low-carbon scrubbing liquid, or</li> <li>Use of lower-strength pipe steel, or</li> <li>Replacement of the corroded, remaining in progress.</li> </ul>	

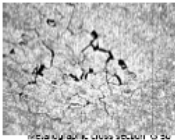
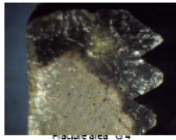
API Recommended Practice 571		December 2003
5.1.2.2.2 <b>MATERIAL DAMAGE (Blistering/CRA/CRA/CRA)</b>		
5.1.2.2.2.1 <b>Description of Damage</b>		
This section describes four types of damage that result in blistering and/or cracking of carbon steel and low alloy steels in wet H <sub>2</sub> S environments.		
41 <b>Hydrogen Blistering</b>		
Hydrogen blisters may form as surface bulges on the ID, the OD or within the wall thickness of a pipe or pressure vessel. This blister results from hydrogen atoms that form during the surface corrosion process on the surface of the steel, that diffuse into the steel, and collect as a discontinuity in the steel such as an inclusion or imperfection. The hydrogen atoms combine to form hydrogen molecules that are too large to diffuse out and the pressure builds in the steel where local deformation occurs, forcing a blister. Blistering results from hydrogen generated by corrosion, not hydrogen gas from the process stream. (Figure 5-10 and Figure 5-11.)		
42 <b>Hydrogen Induced Cracking (HIC)</b>		
Hydrogen blisters can form at many different depths from the surface of the steel. In the middle of the pipe or near a weld, in some cases, engineering or adjacent means that use of slightly different depths, patterns may develop cracks that look from together. Interpenetrating cracks between the blisters often form a hair-like appearance, and so HIC is sometimes referred to as "hair-like cracking." (Figure 5-12, Figure 5-13 and Figure 5-14.)		
43 <b>Stress-Orientation Hydrogen Induced Cracking (SOHIC)</b>		
SOHIC is similar to HIC, but is a generally more damaging form of cracking which appears as cracks in areas of cracks oriented on top of each other. The result is a through thickness crack that is perpendicular to the surface and is driven by high levels of stress residual or applied. They usually occur in the base metal adjacent to the weld heat affected zones where they originate from HIC damage or other cracks or defects including surface stress cracks (Figure 5-15 and Figure 5-17).		

API Recommended Practice 571		December 2003
5-40		
		
Figure 5-10 - Cross section of plate showing HIC damage in the shaft of a trim motor which had been cooling vapors of a H <sub>2</sub> S feed in a hydroprocessing unit.		
		
Figure 5-14 - High magnification photograph of HIC damage.		

Corrosion Atlas, Elsevier

API 571

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CORROSION IN REFINERY INDUSTRY		CORROSION IN REFINERY INDUSTRY	
CASE HISTORY	n° 108 January 2002 FR-IFP-Materials Department	CASE HISTORY	n° 108 January 2002 FR-IFP-Materials Department ANSWER
HYDROGENATION UNIT 217 - HOT SEPARATOR V23 SCREWS CEDI - SOLAIZE - IFP		TYPE OF CORROSION : MOLTEN METAL CORROSION	
DATE OF INCIDENT AND/OR INFORMATION:	24/11/01 SA10	CAUSES :	
NATURE OF THE INCIDENT : temperature breakaway and breaks of screws		X ray analysis have been performed with a Scanning Electron Microscope: *on the fracture area Zinc, Cadmium, Sulphur, Oxygen were detected *In the cracks Cadmium with Oxygen and locally Zinc, Chloride, Copper and Sulphur had been analysed. As temperatures up to 480°C have been reached, Zinc and Cadmium were in the liquid state (the melting temperature of Cadmium is 320°C and the one of Zinc is 420°C) and they can provide a molten metal type corrosion of the carbon steel screws. The blocking stresses have surely favoured the attack by the molten metals	
CONSEQUENCES : stop of the unit		REMEDY :	
MATERIAL: carbon steel		The origin of the zinc and the cadmium are looking for (paint, surface treatment of the screws, grease... ?) in order to avoid them.	
PHOTO AND SCHEME:		PUBLICATION - TECHNICAL REPORT: IFP technical note n°20/2002 RG30	
 		BIBLIOGRAPHIC REFERENCES :	
ASPECT : some parts of the fracture zone have a green colour and the others a dark brown one. The metallographic examination has revealed intergranular cracks.		Typical refinery failure cases atlas ?	
MEDIA AND OPERATING CONDITIONS: Air, normal operating temperature 380°C but increases up to 480°C			
TIME TO DETERIORATION : A few months (design in August 2000)			

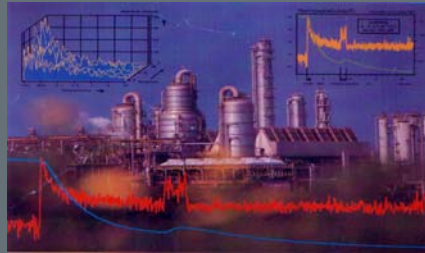
WP 15 meeting September 6<sup>th</sup> 2005 Lisbon

## **Appendix 4**

### **Electrochemical noise in corrosion monitoring**

**A. Cafissi and S. Trasatti (University of Milan)**

# “Electrochemical Noise in corrosion monitoring”

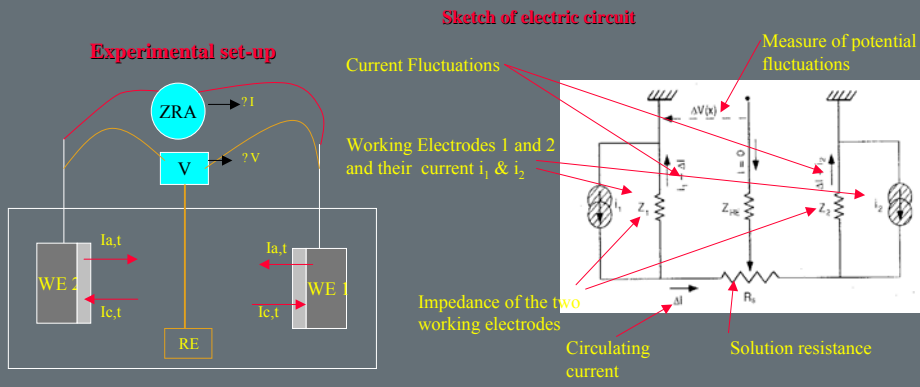


A. Cafissi and S.P. Trasatti  
 Department of Physical Chemistry and Electrochemistry-  
 University of Milan- via Venezian, 21- 20133 MILAN (Italy)

EUROCORR 2005 4-8 September 2005 , Lisboa (P)

## Electrochemical Noise

- Electrochemical technique entails with registration and analysis of potential and current fluctuations associated with electrochemical phenomena at the electrode/solution interface.
- Current fluctuations between two working electrodes involve in the process are in the range of  $10^{-11}$ - $10^{-3}$  A
- Potential fluctuations between the working electrodes and the reference electrode are in the range  $10^{-6}$ - $10^{-1}$  V



## Applications of Electrochemical Noise

The EN has a wide range of applications, i.e.:

- Determination of active/passive transition
- Study of the passive film breakdown (pitting, crevice corrosion, etc.)
- Growth and detachment of hydrogen bubbles (acid corrosion)
- Determination of fluctuations due to diffusive phenomena
- Coating failure on metallic substrate
- Biological corrosion phenomena
- Electrocrystallization associated with diffusion processes

## Analysis of Electrochemical Noise data

It can be performed as follows:

First:

- Direct analysis of raw data

Second

- Statistical analysis:  $d_v$ ,  $d_i$ , variance, kurtosis, skewness, average values, etc.

- Determination of  $R_n$  (Noise resistance), a useful parameter for the corrosion rate:

$$(R_n \approx R_p) \quad R_n = d_v / d_i$$

- Determination of localization index

$$L_I = d_i / I_{rms}$$

Third

- Fourier Transform analysis (power spectral density, PSD).

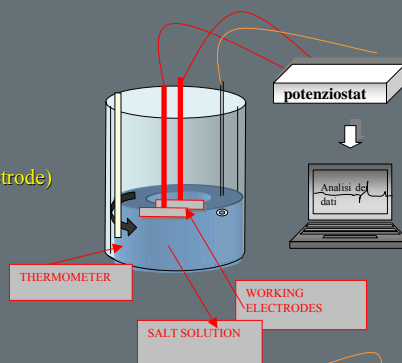
## Experimental scope

### Evaluation of Electrochemical Noise technique for monitoring corrosion of industrial plants

- Study of different corrosion morphologies
- Study of the influence of physical and mechanical parameters (flow rate, bacteria, temperature, tensile load, etc)
- Definition of statistical parameters ( $d_v$ ,  $d_f$ ,  $R_n$ ,  $L_D$ , PSD)
- Development of a statistical methodology of the electrochemical signal by the use of neural networks

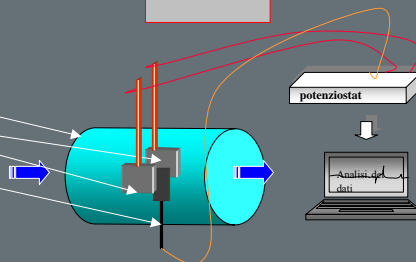
### Acquisition set up for static experiment

- Potenziostat/galvanostat in ZRA configuration
- A reactor for electrochemistry reaction
- A reference electrode (i.e. SCE saturated calomel electrode)
- Two identical working electrodes



### Acquisition set up for dynamic experiment

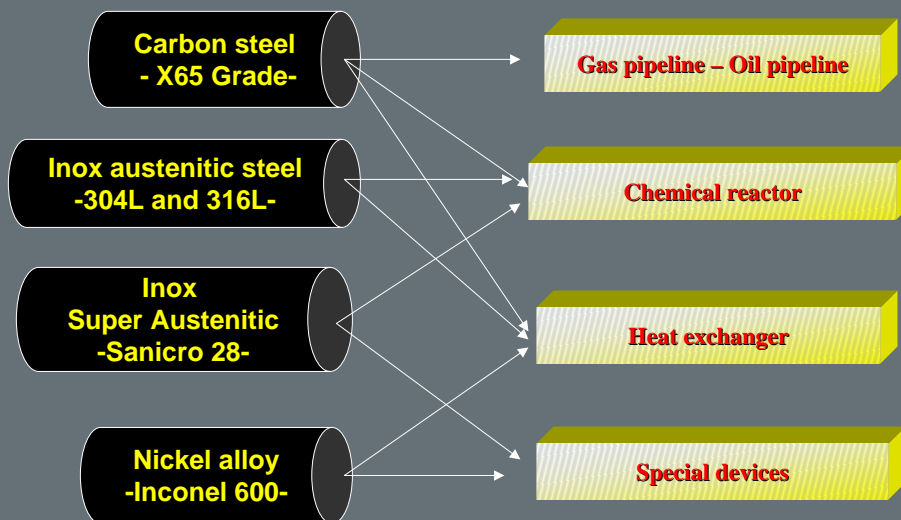
- A flowdynamic reactor
- Two working electrode of steel
- A metallic (Pt,Ru) reference electrode



## Informations available from different testing conditions

<u>Testing</u>		<u>Experiment Purpose</u>
Static test	⇒	Study of different corrosion morphology
Tensile test	⇒	Applicability of EN in stress corrosion cracking
SRB corrosion test	⇒	Monitoring the influence of SRB bacteria
Flow dynamic test	⇒	Monitoring the influence of dynamic conditions

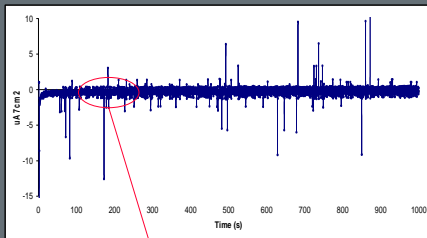
## Materials



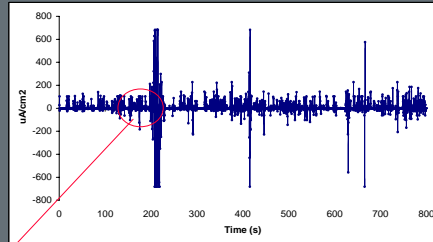
## INFLUENCE OF BACTERIA ACTIVITY ON STEEL CORROSION

- Bacterial activity**
- forms a biofilm on the metallic surface
  - reduces the activity of inhibitors
  - forms corrosive H<sub>2</sub>S gas in solution
  - prevents oxygen diffusion in the solution
  - creates a differential aeration on the substrate

- Bacterial activity causes**
- increase of corrosion current
  - metal depassivation
  - blistering in the steel
  - decrease in pH solution



Current fluctuations for a typical steel pipe in a 3.5% NaCl solution in the absence of SRB bacteria: current spike of 12 μA/cm<sup>2</sup>



Current fluctuations for steel pipe in a 3.5% NaCl solution in the presence of Bacteria SRB (temperature: 30 °C): current spike in the range of 600 μA/cm<sup>2</sup>

Immersion Time (h)	μA/cm <sup>2</sup> X65	μA/cm <sup>2</sup> X65 + SRB	mm/y X65	mm/y X65+SRB
8	5	200	31.4 x 10 <sup>-3</sup>	2.51



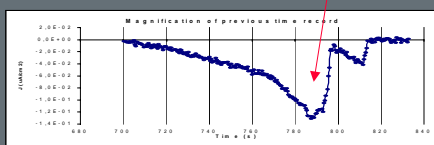
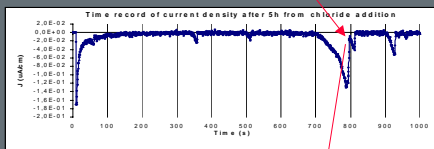
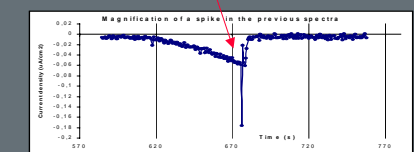
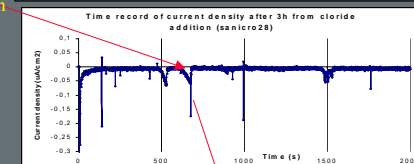
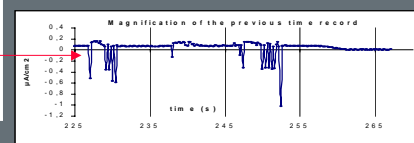
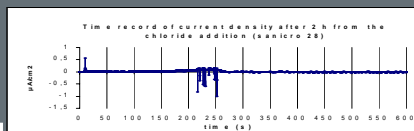
Surface appearance of carbon steel after SRB corrosion

## Super austenitic stainless steel : Sanicro 28

Typical Time record of current density for Sanicro 28 in a 3,5% NaCl solution at room temperature

Immersion Time (h)	μA/cm <sup>2</sup> Background Noise	Spike Charge (μC/cm <sup>2</sup> )	CR mm/y Background Noise
2	0,02	0,3	2,5 x 10 <sup>-4</sup>
3	0,016	1,8	2,0 x 10 <sup>-4</sup>
5	0,01	3,6	1,25 x 10 <sup>-4</sup>

Note: very good material , so very few spikes & low corrosion

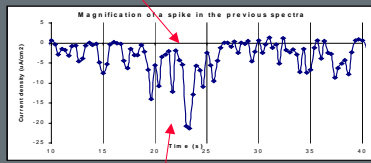
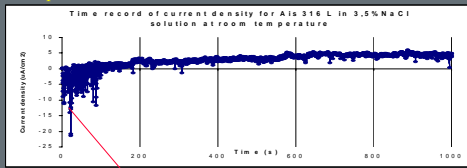


## AISI 316 L

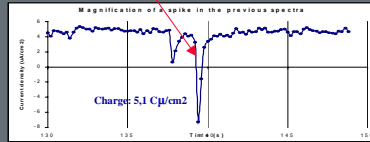
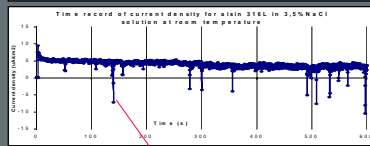
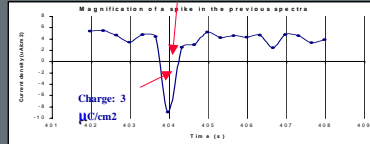
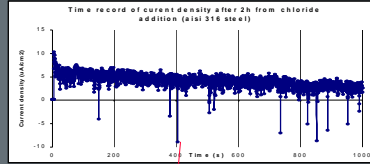
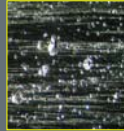
**Time record of current density for AISI 316L Steel in a 3,5% NaCl solution at room temperature and relative magnification of transient spike**

Immersion Time (h)	$\mu\text{A}/\text{cm}^2$ Background Noise	Spike Charge ( $\mu\text{C}/\text{cm}^2$ )	Height of Spike $\mu\text{A}/\text{cm}^2$	CR: mm/y Background Noise
2	2,7	3	8	$33,8 \times 10^{-3}$
3	2	5,1	7	$25,1 \times 10^{-3}$
5	3	31,5	22	$37,6 \times 10^{-3}$

**Note :** Grow of current spikes and localized corrosion as the test proceed



Charge: 31,5  $\mu\text{C}/\text{cm}^2$



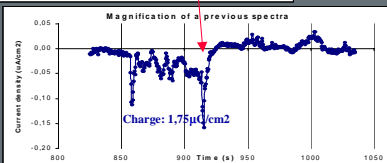
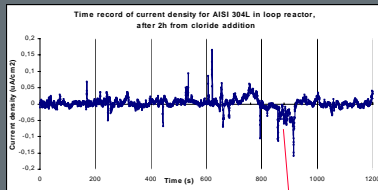
Charge: 5,1  $\mu\text{C}/\text{cm}^2$

## AISI 304 L in the loop

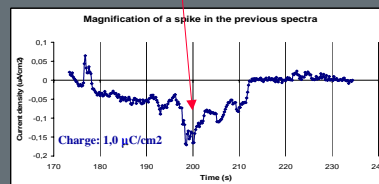
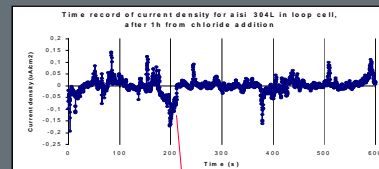
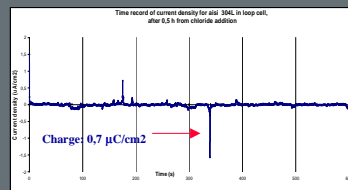
**Time record of current density for Aisi 304L in loop reactor in a 3,5% NaCl solution at room temperature and relative magnification of transient spikes**

**Note :** Very low current density due to high oxygen concentration caused by turbulence conditions (flow rate 6 l/min).

Immersion Time (h)	$\mu\text{A}/\text{cm}^2$ Background Noise	Spike Charge ( $\mu\text{C}/\text{cm}^2$ )	CR: mm/y
2	0,067	0,7	$8,36 \times 10^{-4}$
4	0,03	1	$3,8 \times 10^{-4}$
6	0,01	1,75	$1,36 \times 10^{-4}$



Charge: 1,75  $\mu\text{C}/\text{cm}^2$



Charge: 1,0  $\mu\text{C}/\text{cm}^2$



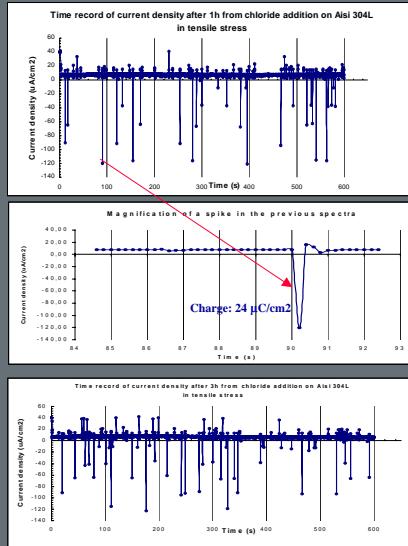
## SCC on AISI 304L

Time record of current density for AISI 304L in a 3,5% NaCl solution at different time and with constant tensile stress, Relative magnification of transient spike

Immersion time Time (h)	Background Noise $\mu\text{A}/\text{cm}^2$	Height spike $\mu\text{A}/\text{cm}^2$	Spike Charge $\mu\text{C}/\text{cm}^2$	CR mm/y
1	7,7	110	24	$9,7 \times 10^{-2}$
3	8	120	27	0,1

Note :

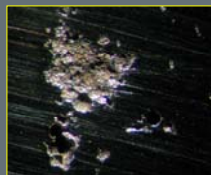
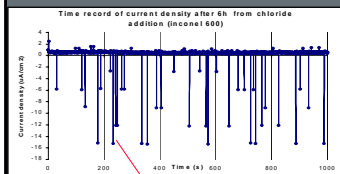
- EN can be very helpful in monitoring the passive film break down (amplitude of the current spike), increased by the tensile stress applied that expose a new not passivated surface.



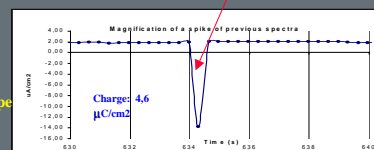
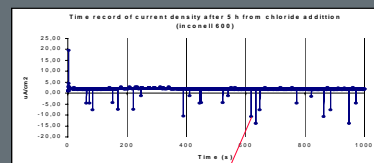
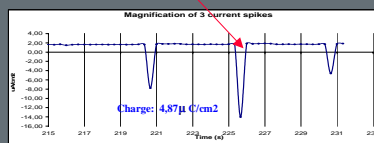
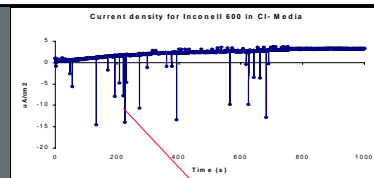
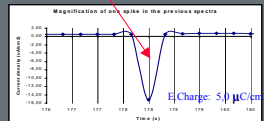
## Inconel 600 alloy

Time record of current density for Inconel 600 in a 3,5% NaCl solution, pH=3, at room temperature and relative magnification of transient spike

Immersion Time (h)	Background Noise $\mu\text{A}/\text{cm}^2$	Spike Charge $\mu\text{C}/\text{cm}^2$	Height of Spike $\mu\text{A}/\text{cm}^2$	mm/y Background Noise
3	6	4,9	15	$7,5 \times 10^{-2}$
5	1,4	4,6	14	$1,8 \times 10^{-2}$
6	0,5	5	15	$0,63 \times 10^{-2}$



Photograf taken by optical microscope



## Noise spectra transformation from the time domain to the frequency domain (FFT to obtain the PSD)

The aim of this transformation is:

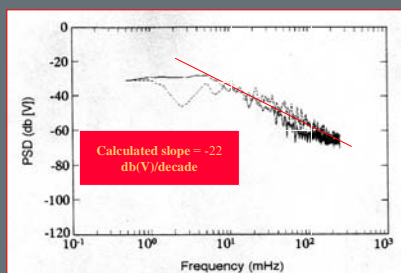


To filter the frequencies not associated to the electrochemical reactions (amplification noises, instrumentation interferences, power current).

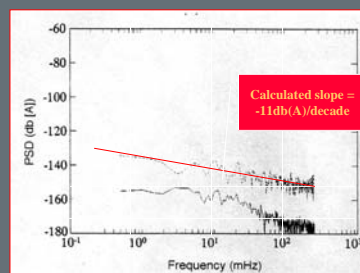


To determine the slope of the (PSD) spectra as an useful parameter to distinguish between uniform and localized corrosion.

## Potential and current PSD Localized corrosion



PSD (V) for localized corrosion



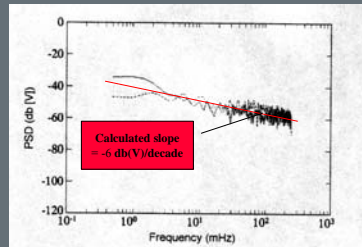
PSD (I) for localized corrosion

Type of Corrosion	Slope (db[V]/decade)		Slope (db[A]/decade)	
	Max.	Min.	Max.	Min.
Passivation	-15	-25	1	-1
Pitting	-20	-25	-7	-14
Uniform	0	-7	0	-7

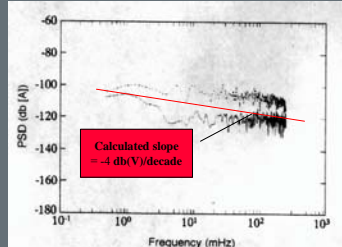
Slope values for different corrosion modes

## Potential and current PSD

### General corrosion



PSD (V) for uniform corrosion



PSD (I) for uniform corrosion

Type of Corrosion	Slope (db(V)/decade)		Slope (db(A)/decade)	
	Max.	Min.	Max.	Min.
Passivation	-15	-25	1	-1
Pitting	-20	-25	-7	-14
Uniform	0	-7	0	-7

Slope values for different corrosion modes

## Theoretical calculation of metal mass loss during a current spike

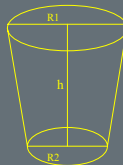
### Determination of hypothetical Pit volume Metal Dissolution during a Current Spike

IF  $R_1$  = radius maior of Pit in  $\mu\text{m}$   
 IF  $R_2$  = radius minus of Pit in  $\mu\text{m}$   
 IF  $h$  = depth of the pit hole in  $\mu\text{m}$   
 IF  $\gamma_{\text{Fe}} = 7.3 \mu\text{g}/\mu\text{m}^2$   
 We obtain :

$$\rightarrow V_{\text{pit}} = 1/3 \pi h (R_1^2 + R_2^2 + R_1 \cdot R_2)$$

Example 1:  
 depth  $h = 0.5 \text{ mm} = 500 \mu\text{m}$   
 $\rightarrow R_1 = 4 \mu\text{m} \rightarrow R_2 = 1 \mu\text{m}$   
 $V_{\text{pit}} \approx 11.5 \times 10^3 \mu\text{m}^3$   
 $\mu\text{g}_{\text{Fe}} = 84.1 \times 10^{23} \mu\text{g Fe} = 84.1 \text{ mg}_{\text{Fe}}$

Example 2:  
 depth  $h = 0.05 \text{ mm} = 50 \mu\text{m}$   
 $\rightarrow R_1 = 4 \mu\text{m} \rightarrow R_2 = 1 \mu\text{m}$   
 $V_{\text{pit}} \approx 11.5 \times 10^3 \mu\text{m}^3$   
 $\mu\text{g}_{\text{Fe}} = 8.41 \times 10^{23} \mu\text{g Fe} = 8.41 \text{ mg}_{\text{Fe}}$



Theory:

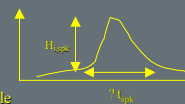
IF  $H_{\text{1,spk}}$  = Height of spike i in  $\mu\text{A}$   
 IF  $? t_{\text{1,spk}}$  = duration time of a single spike i  
 IF  $M_{\text{eq}}$  = equivalent weight of Fe :  $M_{\text{eq}} = 28 \text{ g/mole}$   
 IF  $F = 96500 \text{ C}$  (electric charge of 1mole of electrons)  
 IF  $f_{\text{1,spk}}$  : frequency of spike event :  $n^{\circ}\text{spike/sec}$

What's the amount of metal during a dissolution spike ...  $m_{\text{Fe}}$ ?

$$m_{\text{Fe}} = (H_{\text{1,spk}} \cdot ? t_{\text{1,spk}} \cdot M_{\text{eq}}) / F$$

Example 1:  
 $H_{\text{1,spk}} = 10 \mu\text{A}$ ;  $? t_{\text{1,spk}} = 0.6 \text{ s}$   
 $m_{\text{Fe}} = 1.74 \cdot 10^3 \text{ g Fe}$

Example 2:  
 $H_{\text{1,spk}} = 100 \mu\text{A}$ ;  $\Delta t_{\text{1,spk}} = 0.6 \text{ s}$   
 $m_{\text{Fe}} = 1.74 \cdot 10^3 \text{ g Fe}$



metal dissolution during a day ?  $m_{\text{Fe (day)}}$ ?

$m_{\text{Fe}} = 2.51 \cdot 10^3 \text{ g Fe}$        $m_{\text{Fe}} = 2.51 \cdot 10^4 \text{ g Fe}$

Mass of Iron dissolved in general corrosion:  $m_{\text{general}}$

IF  $I_{\text{corr}}$  : Background Current Noise :  $2 \mu\text{A/cm}^2$

$M_{\text{general}} : 5 \cdot 10^3 \text{ g}_{\text{Fe}}/\text{day cm}^2 \rightarrow 0.1 \mu\text{m}/\text{day cm}^2$

## Practical examples

For:

- a mean height spike :  $H_{i,spk} = 15, 30, 60$  or  $100 \mu A/cm^2$
- a mean duration time of the spike ?  $t_{spk} = 3$  seconds
- a mean number of spike in a day equal to 1440 or 4320 spikes
- and we assume a hypotetic pit , with this characteristics:  $Depth = 25 \mu m$  ,  $V_{pit} \approx 5,76 \times 10^{-4} \mu m^3$  ,  $4,2 \cdot 10^{-4} g Fe$

We obtain the pit formation for 1440 spike /day in this time:

Mean Height Spike ( $\mu A/cm^2$ )	Relative Mass of Iron dissolved (g)	Time to pit (Days)
15	$1,88 \times 10^{-9}$	223
30	$3,76 \times 10^{-9}$	112
60	$7,52 \times 10^{-9}$	56
100	$1,26 \times 10^{-8}$	33,4

And the pit formation for 4320 spike /day in this time:

Mean Height Spike ( $\mu A/cm^2$ )	Relative Mass of Iron dissolved (g)	Time to pit (Days)
15	$5,64 \times 10^{-9}$	74,3
30	$11,28 \times 10^{-9}$	37,3
60	$22,56 \times 10^{-9}$	18,6
100	$3,78 \times 10^{-8}$	11,1

## CONCLUSIONS

- Electrochemical Noise analysis can be successful applied to investigate corrosion processes and to estimate corrosion rate
- Testing in the presence of SRB activity has shown that the corrosion onset and propagation growth is strictly connected to the bacteria metabolism.
- EN can be conducted under both static and dynamic conditions as well as under tensile loading
- The PSD slope can be a useful parameter to distinguish among different corrosion modes

## **Appendix 5**

# **New Technology for Prediction and Assessment of Corrosion in Refinery Operations**

**R.D. Kane (Honeywell)**



# New Technology for Prediction and Assessment of Corrosion in Refinery Operations

Russell D. Kane  
InterCorr International, Inc.\*

\*Recently acquired by Honeywell

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## Organization

- Topics of Discussion
  - Ammonium Bisulfide Corrosion
    - SourWater JIP Phase I and II
    - Predict®-SourWater Software; release 2.0
  - Amine Unit Corrosion
    - Amine JIP
    - Predict®-Amine
  - New JIP Activities starting in 2006
    - Sulfuric Acid Alkylation
    - High Temperature Crude Oil Corrosivity - II

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2

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## SourWater JIP Phase I & II

- Title: "Prediction & Assessment of Ammonium Bisulfide Corrosion Under Refinery Sour Water Service Conditions" Phase I and II
- Jointly developed by InterCorr and Shell Global Solutions
- Phase I – H<sub>2</sub>S-dominated sour water systems (REAC)
  - Two year effort completed in 2003
    - **Extensive engineering database** – parametric study of NH<sub>4</sub>HS concentration, H<sub>2</sub>S partial pressure, temperature, inhibitors, flow (wall shear stress).
    - **17 sponsors / approx. \$1 million US**
    - **Predict®-SW Software** – an integrated software tool for corrosion prediction and flow modeling for use in materials selection, risk assessment, inspection planning and enhanced unit operation.
    - **Predict-SW 2.0 Software** – is now available to both sponsors and non-sponsors; single-user and multi-user corporate licenses.

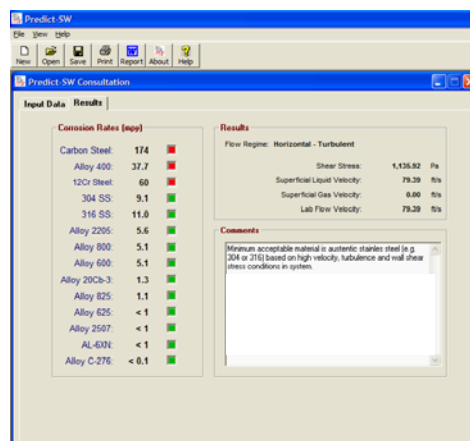
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## Predict®-SW 2.0

- Use of design or actual unit conditions to assess of corrosion severity. Includes **H<sub>2</sub>S partial pressure effects** not found in current methods (e.g. K<sub>p</sub>).
- Provides multiphase **flow modeling** to link unit conditions with JIP data.
- Queries **iso-corrosion and parametric relationships** over a broad range of system conditions.
- Output provides predicted corrosion rates for specific alloys based on
  - environmental conditions
  - flow regime, and
  - wall shear stress.
- Release 2.0 includes new features:
  - Updated Rules / Sensitivity Function
  - View Data
  - Units compatible with outputs from process models
  - Benchmarking, reporting & online Help

Output Screen



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## SourWater JIP – Phase II

### Phase II – NH<sub>3</sub>-dominated sour water systems

- Phase II currently in progress.
- Extensive engineering database – parametric study of NH<sub>4</sub>S concentration, **H<sub>2</sub>S & NH<sub>3</sub> partial pressures**, temperature, **cyanides**, flow (wall shear stress).
- Involved 11 companies and approx. \$700,000 in funding.
- New Information: **Conclusively shows a reversion to higher corrosion rates at high ammonia partial pressures.**
- Predict-SW software will be updated with this new information/rules.
- Program sponsors have two year confidentiality period.
- JIP will run through mid-2006.

### Sour Water Participants (I & II)

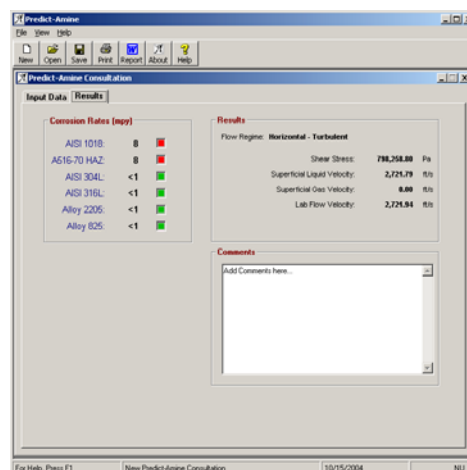
- ChevronTexaco Energy Research & Technology Co.
- Flint Hills Resources, L.P
- Shell Global Solutions (US) Inc.
- ConocoPhillips, Inc.
- Petrobras - CENPES
- ExxonMobil Research & Engineering
- Sunoco
- Fluor Daniel Inc.
- UOP
- Saudi Aramco
- Syncrude Canada Ltd.
- Phillips Petroleum Co.
- Valero
- TOTAL
- BP
- Kuwait National Petroleum Co.
- Idemitsu Kosan Co., Ltd
- Lyondell-Citgo Refining LP
- Marathon-Ashland Petroleum

## Amine JIP

### Minimizing Amine Unit Corrosion

- Data development combining chemical and flow simulation (similar to Sour Water program)
- Program initiated with assistance of Flint Hill Resources.
- The program involves five tasks:
  - Task 1 – Baseline Data for Sour MEA, DGA, DEA Systems
  - Task 2 – Temperature
  - Task 3 – CO<sub>2</sub>/H<sub>2</sub>S Ratio
  - Task 4 – Heat Stable Salts
  - Task 5 – Development of a Software Corrosion Prediction Tool (Predict-Amine)
- Recently, tests included to evaluate MDEA.
- Program to be complete by mid-2006.
- Single user software license and 2 yr confidentiality period.
- New Lean Amine JIP being formulated.

### Output Screen





## 2006 JIP Activities

- The previous JIP's indicated that much work is needed to quantify the effects of process chemistry and flow conditions on corrosion of commonly used alloys.
- This approach has proven successful in studying:
  - Nap acid, sour water and amine corrosion.
- Based on end-user input, a similar parametric approach is being applied to:
  - Sulfuric acid alkylation unit corrosion
  - Current Nap Acid issues related to present limits for carbon steel, 5Cr, 9Cr, 12Cr and 316/317
- Programs are scheduled to begin in 2006:
  - Sulfuric acid alkylation: Q1 2006
  - Nap acid: Q3 2006

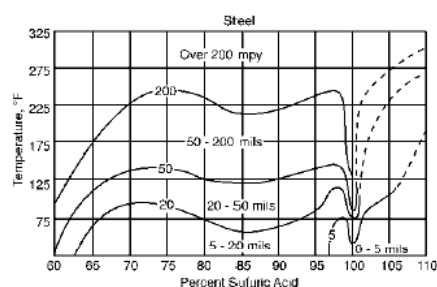
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## Sulfuric Acid Alkylation JIP - 1

### Prediction of Corrosion in Sulfuric Acid Alkylation Units

- A highly profitability refining operation.
- There is pressure on to maximize productivity (thru-put & availability) of alkylate processing units.
- Program was based on a gaps analysis conducted in early 2005:
  - Corrosion rates for steel at low temperatures
  - Influence of ASO
  - Role of contaminants
  - Limits for 316 and Alloy 20 at high temperatures
  - Influence of welding
  - Influence of flow & regime

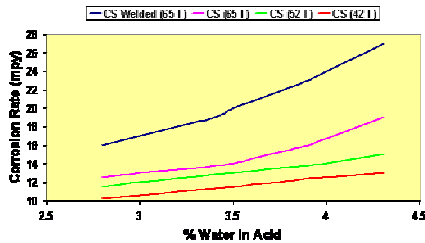


Limited base data in API 581 for reagent grade  $H_2SO_4$  corrosion from 1950's

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## Sulfuric Acid Alkylation JIP - 2



Limited data developed more recently indicates an effect of welding, temperature and velocity, but does not take into account wall shear stress.

- **JIP Technical Program:**
- **Task 1** – Chemical & Flow Modeling
- **Task 2** – Sensitivity studies on spent acid concentration, ASO and oxygenates
- **Task 3** – Parametric Study
  - Acid concentration
  - Temperature
  - Wall shear stress
- **Task 4** – Corrosion modeling and development of Predict®-SA software.
- Sponsors will have a 2 year confidentiality period
  - Includes single user version of software and use of data

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## Minimizing Refinery Crude Corrosivity II

- Phase I was a pioneering JIP
  - Included support from over 20 companies
  - the most comprehensive experimental study.
- It identified critical parameters needed to better characterize the corrosivity of crude oil systems.
- Mechanisms of Nap Acid corrosion.
- Quantification of both chemical and flow-related mechanical factors.
- Limits for impingement attack of 5Cr, 9Cr & 12Cr.
- Influence of Nap Acid ring structures and sulfur speciation on corrosivity.
- Crude Corrosivity JIP-II starts in mid-2006
- Sponsors will have a 2 year confidentiality period
  - Includes single user version of software and use of data



- New program focuses on:
  - Major limitations of currently utilized risk matrix in API 581
  - Combined effects of TAN and sulfur
  - Liquid filled lines in turbulent flow for carbons steel, 5Cr, 9Cr and higher alloys
  - Needs for quick screening test for corrosivity assessment
  - Corrosivity software model (Predict®-Crude) to work with in-house proprietary assessment tools.

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## Summary

- New Technology for Prediction and Assessment of Corrosion in Refinery Operations:
  - Ammonium Bisulfide Corrosion
    - SourWater JIP Phase I and II
    - Predict®-SourWater Software; release 2.0
  - Amine Unit Corrosion
    - Amine JIP
    - Predict®-Amine
  - New JIP Activities starting in 2006
    - Sulfuric Acid Alkylation
    - High Temperature Crude Oil Corrosivity – II

For more information contact R.D. Kane at: [russ.kane@honeywell.com](mailto:russ.kane@honeywell.com)

## **Appendix 6**

### **Nickel alloys in hydroprocessing units**

**J. Hucinska (Gdansk University)**

# Nickel alloys in hydroprocessing units

JOANNA HUCIŃSKA

Gdańsk University of Technology, Gdańsk, Poland

EUROCORR 2005, EFC WP15

September 6, 2005 • Instituto Superior Tecnico • Lisbon, Portugal

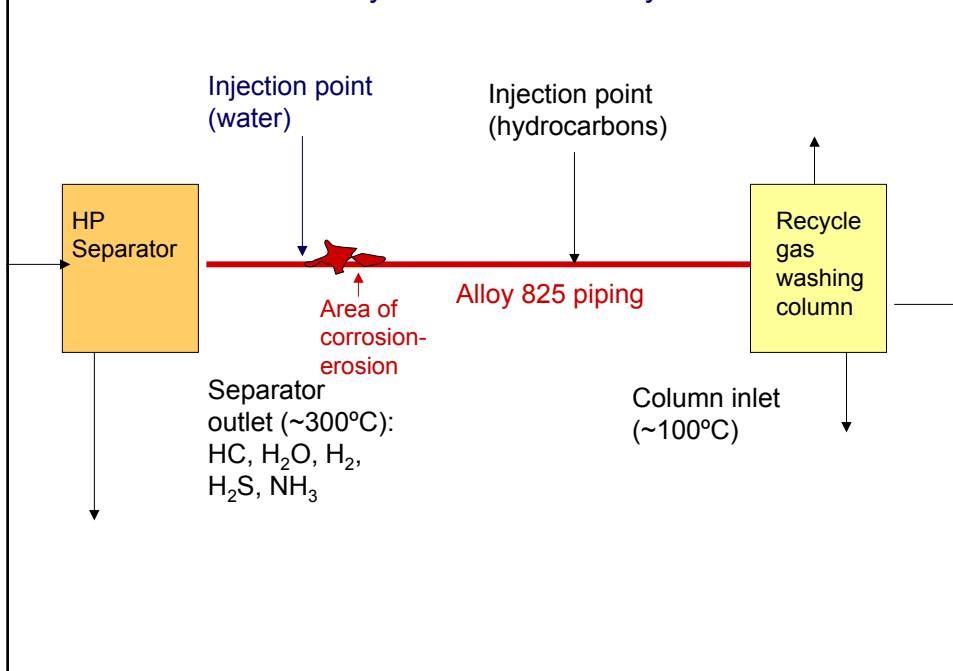
## Lay-out

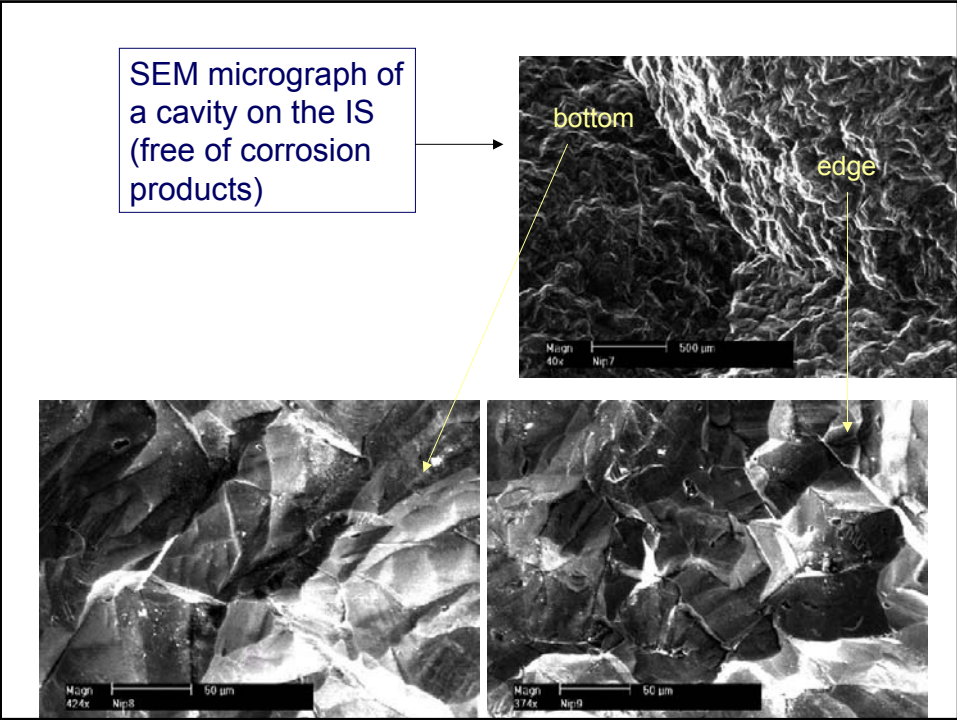
- Corrosion-erosion of reactor effluent air coolers (REACs) and piping
- Corrosion-erosion of Alloy 825
- Questions

## Corrosion-erosion of reactor effluent air coolers (REACs) and piping

- Problem in hydrocracking and high severity hydrotreating units
- Reasons of damage: presence of aqueous ammonium bisulphide ( $\text{NH}_4\text{HS}$ ) and multiphase flow conditions
- Materials of construction: carbon steel, 321 steel, Alloy 800, Alloy 825

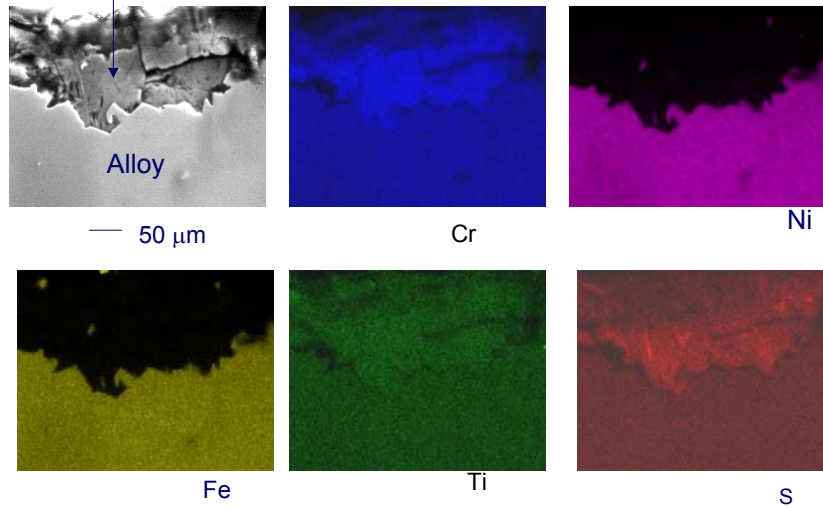
## Corrosion-erosion of Alloy 825 in Used Oil Hydrotreatment Plant





## Microscopic characteristics of corrosion-erosion

Corrosion products



Corrosion products on the IS, cross-section. SEM, EDS

## Reasons of Alloy 825 brittle fracture

- $(\text{NH}_4\text{HS})$  + cavitation/droplet impingement
- Hydrogen?
- ?



## **Appendix 7**

### **Metal dusting in plattforming CCR furnaces**

**J. Hucinska (Gdansk University)**

# Metal dusting in Platforming CCR furnaces

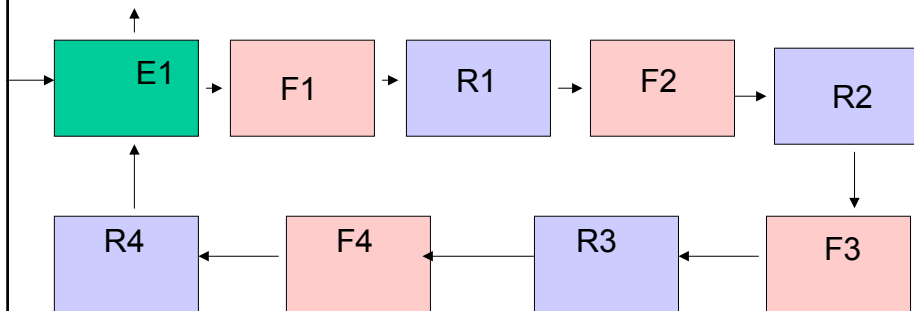
JOANNA HUCIŃSKA  
Gdańsk University of Technology, Gdańsk, Poland

EUROCORR 2005, EFC WP15  
September 6, 2005 • Instituto Superior Tecnico • Lisbon, Portugal

## Lay-out

- Characteristics of furnaces, tubes, service conditions
- Non-destructive testing
- Destructive testing
- Questions

### Unit: Platforming CCR (continuous catalytic reforming)



### Characteristics of tubes and service conditions

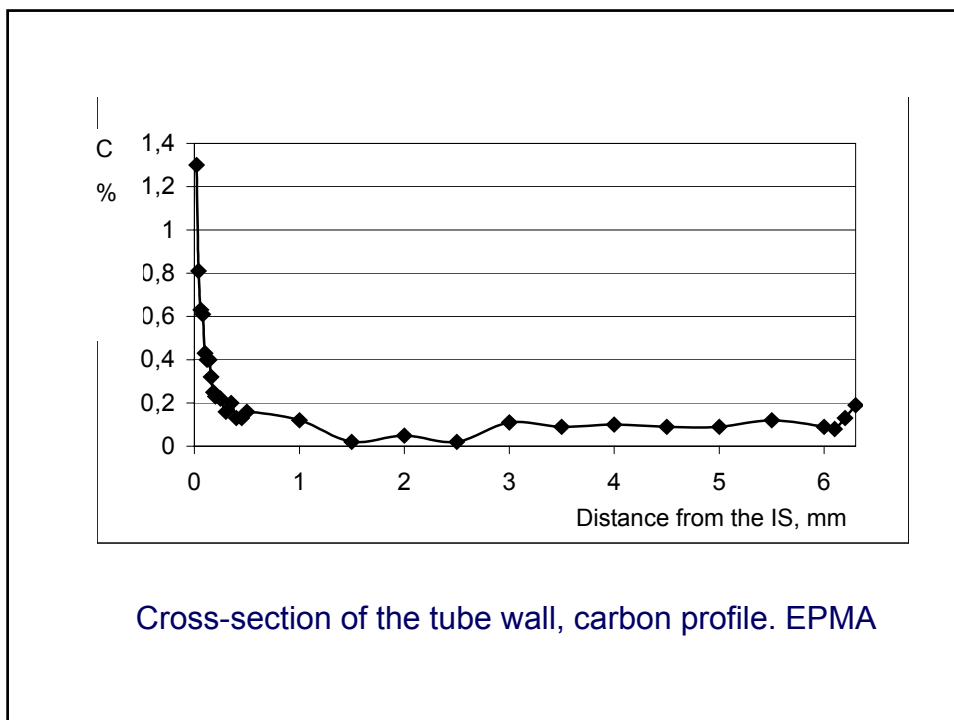
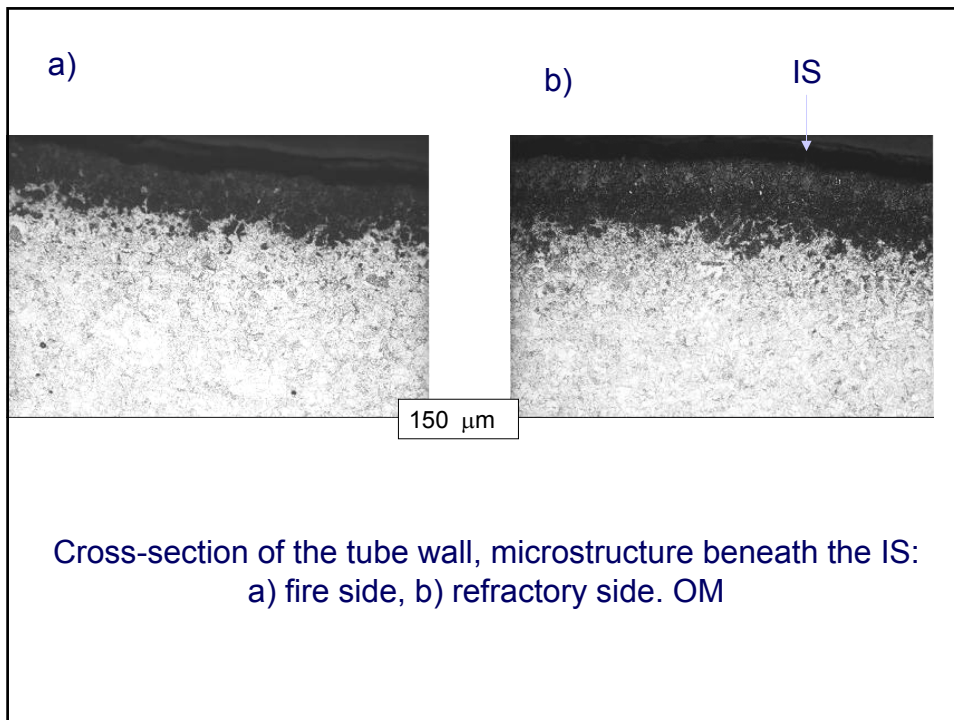
Characteristics	Description
Material of tubes	Steel 9Cr-1Mo, gr P9 ASTM SA-335
Nominal dimensions	114,3 x 6,0 mm
Environment	Heavy naphtha after desulphurisation and hydrogen gas: 440-520°C (F1), 400-520°C (F2), 450-520°C (F3), 470-520°C (F4), S~0.2 ppm (naphtha), S<0.5 ppm (hydrogen), ppH <sub>2</sub> ~0.65-0.8 MPa
Metal temperature	635°C (design), ~600 °C (service)
Time of service	10 years

## Non-destructive testing

- Visual – no scale on the outside surface (OS) of the tubes
- Wall thickness – no losses
- Ultrasonic wave attenuation measurements – decrease in attenuation

## Destructive testing

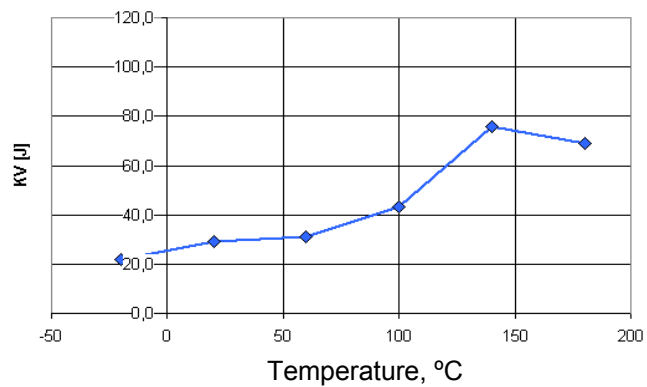
- Visual – thin, loose deposits of carbon on the inside surface (IS) of the tubes
- Metallographic: optical microscope (OM), transmission electron microscope (TEM)
- Electron probe microanalyser (EPMA)
- Tensile tests, Charpy-V impact tests

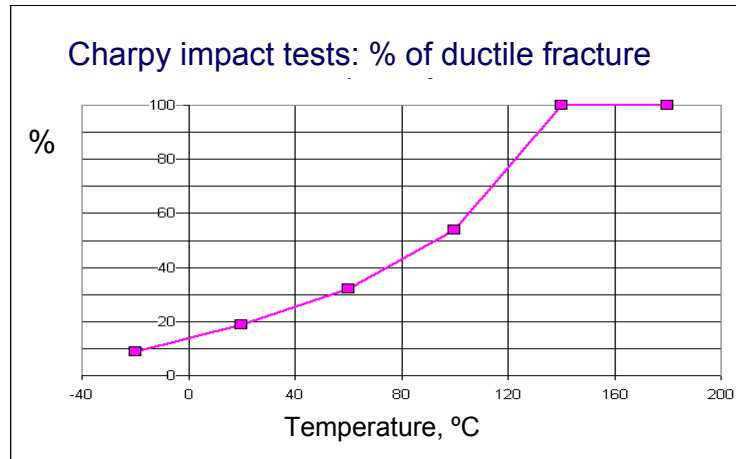


## Tensile tests

Material	Yield Stress MPa	Tensile strength MPa	Elongation %
Before service	337-400	556-588	32
After service	269-281	541-558	20
ASTM SA-335	205 min	415 min	27

Charpy-V impact tests: fracture energy KV



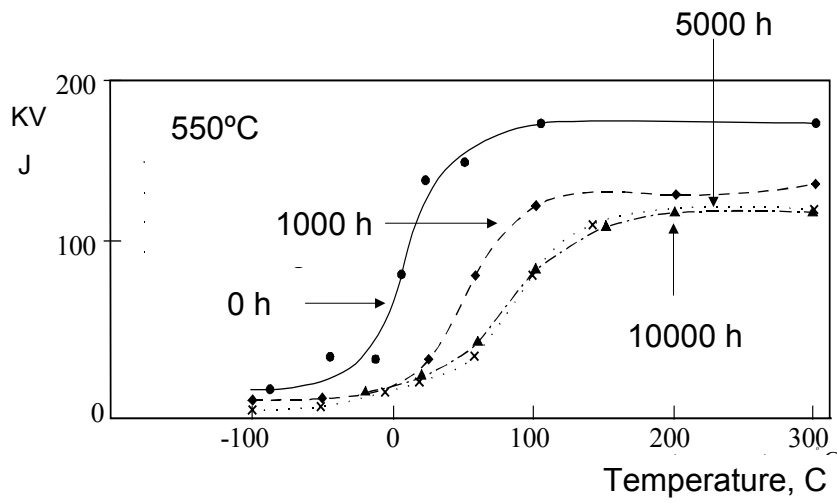


FATT 50=93°C

FATT – fracture appearance transition temperature

Reasons of brittleness?

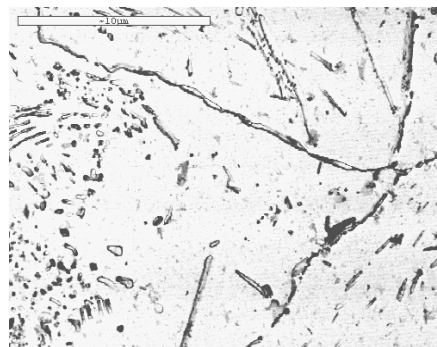
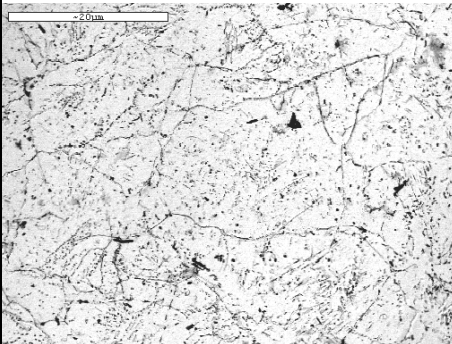
### Literature data



Reason of brittleness: Laves phase  $\text{Mo}_2\text{Fe}$  at grain boundaries

(J.A. Hudson, S.G. Druce, G. Gage, M. Wall: Theoretical and Applied Fracture Mechanics 1988, 10, p. 123)

### Examinations



Mid-wall, carbon extraction replica. TEM



### Chemical composition of precipitates, wt. %

No	Si	P	S	Cr	Mn	Fe	Mo
1	14.05	0.77	0.49	7.78	2.21	33.08	41.61
2	57.41	0.75	-	7.35	-	17.32	17.17
3	6.20	-	-	56.39	1.00	28.26	8.15
4	20.93	-	-	16.59	-	56.01	6.47
5	3.80	0.29	0.37	56.93	-	28.38	10.23
6	23.38	-	-	29.72	-	27.73	19.17
7	16.15	-	-	43.07	-	30.59	10.19

### Questions

- Metal temperature 600°C: too high for 9Cr-1Mo steel?
- Sulphur compounds added to the feed in platforming CCR unit are harmful or advantageous ?